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## Sulfite activation changes character of ferrate resultant particles

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# **Sulfite Activation Changes Character of Ferrate Resultant Particles**

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## **Graphical Abstract**



## **HIGHLIGHTS**

- Sulfite activation results in less magnetic resultant iron particles
- Activation further influences particle morphology and size distribution
- Sulfite activation may impede downstream physiochemical water treatment processes

## **ABSTRACT**

The activation of ferrate with sulfite increases oxidative transformation of recalcitrant organic compounds; however, it also changes the characteristics of the iron particulates that result from the ferrate reduction. In this study, particles resulting from ferrate reduction both with and without sulfite were compared in a laboratory matrix simulating water treatment conditions at the bench-scale. Characteristics examined included magnetization, morphology, size, and surface charge. The activation of ferrate with sulfite changed the characteristics of resultant particles in several important ways. Activated ferrate resultant particles were less magnetic, more polydisperse including a higher fraction of nanoparticles, and exhibited a less-crystalline morphology compared to particles resulting from ferrate self-decay. Surface charges between the two particle types were similar, and negative. The relatively rapid formation of Fe(III) from Fe(VI) activation leads to particles of different character, likely though a greater supply of precursory low molecular weight iron hydroxo-species. Particles resulting from activated ferrate used as a preoxidant will impact downstream processes in important

ways, such as gravimetric or magnetic separations and contaminant adsorption. Ferrate activation presents a possible trade-off between improved oxidation and impeded downstream physicochemical processes, and formation and fate of formed particles warrants consideration.

#### **KEYWORDS**

Activated Ferrate; Iron nanoparticles; Magnetism; Pre-oxidation; Coagulation; Water treatment.

### **1. INTRODUCTION**

Ferrate (Fe(VI)) is a high-valance iron species that is used for oxidative transformation of target compounds.1,2 In a water treatment context, the relatively high oxidation potential of Fe(VI) successfully transforms many inorganic3–5 and organic contaminants,  $6-8$  as well as disinfection byproduct precursors.  $9-12$  However, some recalcitrant contaminants of emerging concern are not effectively oxidized by Fe(VI) at relevant dosages and pH values.13 Oxidation with Fe(VI) can be improved by "activation" of Fe(VI) by common chemical reducing agents,14 acids,15 carbon nanotubes, ammonia,16 and ultraviolet light.17 Disagreement in the literature exists with respect to exact mechanisms of activation, and vary with activation approach; however, an important role of ephemeral iron species (e.g. Fe(V) and Fe(IV)) is likely.

When sulfite is used as the activating agent, the formation of sulfate and hydroxyl radicals also increases oxidation,18 and most published research on activated Fe(VI) has focused on this approach.19,20 For example, oxidation of benzotriazole, phenol, ciprofloxacin and sulfamethoxazole all increased from less than 10% to greater than 75% when 50  $\mu$ M Fe(VI) was activated with 250  $\mu$ M of sulfite (1:4 molar ratio).<sup>21</sup> In this way activated Fe(VI) represents an emerging advanced oxidation technology for enhanced degradation of organic contaminants.18

Prior work on Fe(VI) activated with sulfite (and other reductants) provides important data towards potential adaptation; however, most prior work has been executed in a phosphate buffer to sequester Fe(III) solids that result from Fe(VI) reduction. This dramatically simplifies required analytical steps but prohibits any assessment of resultant particles,<sup>22</sup> which blocks vertical advancement of sulfite-activated Fe(VI). Particles resulting from Fe(VI) self-decay (e.g. non-activated) have shown unique and important properties, including a core-shell architecture,<sup>23</sup> participation in adsorption reactions3,24 and coagulation,10,25,26 magnetism,27 relatively small size,28 and poor settleability.<sup>29</sup> It is yet unknown how reduction of Fe(VI) with sulfite may impact these particle characteristics. The overarching objective of this work was to fill this urgent research gap with an assessment of Fe(III) particles resulting from sulfite activated Fe(VI). Specifically, the characterization included an assessment of magnetism, morphology, size, and surface charge.

#### **2. METHODS AND MATERIALS**

**2.1. Particle Formation Reaction.** High-purity potassium ferrate  $(K_2FeO_4)$ (Element 26 Technology, League City, TX) was added to reagent grade water buffered with 1 mM carbonate at pH 7.5. The Fe(VI) dose was 100  $\mu$ M (5.6 mg/L as Fe), confirmed via absorption at 510 nm.<sup>30</sup> The 1 liter reactors were mixed vigorously  $(G >$ 200 sec<sup>-1</sup>) for one minute, then gradually ( $G \sim 50$  sec<sup>-1</sup>). Water matrix (pH and buffer capacity) and dosing conditions were set to replicate a plausible, low organic carbon water treatment scenario. In Fe(VI) auto-decay experiments, complete Fe(VI) autodecay was confirmed via ABTS method,31 following 60 minutes reaction time. In sulfite activation experiments, 400  $\mu$ M SO<sub>3</sub><sup>-2</sup> as a 0.25 M Na<sub>2</sub>SO<sub>3</sub> stock solution was added to the reactor 30 seconds after Fe(VI), following a similar protocol from prior research focused on activate Fe(VI) oxidation.18

**2.2. Particle Characterizations.** Resulting particle size and surface charge were quantified by dynamic light scattering (DLS) and electrophoretic mobility (Malvern Zetasizer Nano ZS). Particle morphology was imaged with transmission electron microscopy (TEM) (JEM-2100, JOEL, Tokyo, Japan). Resulting particle suspensions for TEM were filtered through a glass-fiber (GF) filter (Whatman, 934AH), with an effective cut-off of 1.5 µm to remove large aggregates. Particles in GF filtrate were then loaded onto a 30 kDa ultrafilter (UF). The UF was submerged and sonicated in reagent grade water, with resuspended particles then drop-casted and air dried on a C film with Cu grid. Particles analyzed by scanning electron microscopy (SEM) (Zeiss Sigma VP, Overkochen, Germany) were prepared by drop casting the as-prepared particle suspension. Crystallinity was evaluated via X-ray powder diffraction (XRD) (Ultima IV, Rigaku, Tokyo, Japan), with a Cu K<sub> $\alpha$ </sub> source at 0.5 deg/min scan speed. The as-prepared ferrate resultant particles were collected on a 0.7 µm cut-off glass fiber fine (GF/F) filter and transferred to the XRD holder for analysis. A sample of GF/F effluent was also subjected to DLS measurements.

**2.3. Evaluation of Magnetism.** Magnetism was assessed using *in situ* and *ex situ*  approaches. *In situ* assessment included transfer of particle suspension to a 10 mm spectrophotometer cell. One, 9.5 mm diameter N40 grade neodymium-iron-boron disc 1 T magnet was affixed to the bottom of the cell on the outside. The apparatus was placed

in a spectrophotometer, and optical density (abs. at 600 nm) was tracked over 30 minutes; the optical density over time was normalized to the optical density at the start of the settling experiment. Experiments were repeated with and without the magnet. *Ex situ* magnetization was measured using a magnetic properties measurement system (MPMS-3 from Quantum Designs, San Diego, USA). 1.5 L of Fe(VI) particle suspension was centrifuged at 4200 g. The supernatant was removed and the pellet was dried in an oven at 80 ºC for 24 h. Approximately 20 mg of powder sample was loaded into the magnetometer and a full cycle from  $-5$  T to  $+5$  T was acquired in DC mode at 26.85 °C.

**2.4. Statistical and Other Information.** Particle formation reactions and subsequent characterizations were conducted in triplicate unless otherwise noted. Graphical error bars represent one standard deviation. More statistical and experimental information is included in Supplementary Information (see SI-S1).

#### **3. RESULTS**

**3.1. Activation and Magnetism**. Fe(VI) decay without sulfite addition (nonactivated ferrate resultant particles, NFRPs) resulted in particles that were separated from fluid by a magnetic field more so than by gravity alone (Figure 1A). After 30 min, the optical densities of the NFRP suspension without and with a magnet were 0.92 and 0.86, respectively. For activated Fe(VI) resultant particles (AFRPs) the optical density both with and without a magnetic were > 0.96, indicating decreased gravity settling, and negligible impact of a magnetic field (Figure 1B). Results for AFRPs were similar to ferric chloride coagulant control (Figure S1). Both particle types settled poorly, a known attribute of Fe(VI) resultant particles;<sup>29</sup> however, only nonactivated particles showed appreciable increase in separation due to a magnetic field.



**Figure 1.** Normalized optical density (absorbance at 600 nm) of particle suspensions resulting from (A) Nonactivated ferrate auto-decay and (B) ferrate activated with sulfite.  $pH = 7.5$ , Fe(VI) dose = 100 µM, 1 mM HCO<sub>3</sub>, 400 µM SO<sub>4</sub><sup>-2</sup>.

Figure 2 shows the mass-normalized magnetization (M) versus magnetic field strength (M(H)) curves near room temperature for both particle types. The nonactivated resultant particles had higher M than AFRPs. This indicates that the portion of ferrimagnetic materials is higher in the nonactivated particles. Additionally, magnetization at high field ( $> 3$  T,) follows a positive, linear trend with H, with similar slopes (0.07 vs. 0.05 emu/g•T), characteristic of paramagnetic materials. Insert B in Figure 2 includes M(H) cycles after paramagnetic subtraction. Results are consistent with prior work demonstrating that nonactivated Fe(VI) resultant

particles exhibit core-shell architecture with a  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> (maghemite; ferrimagnetic) core

and a  $\gamma$ -FeOOH (lepidocrocite; paramagnetic) shell.<sup>23,27</sup> It is important to note, however, that sulfite activation changes the resulting particle architecture by decreasing the ferrimagnetic component.

By subtracting the paramagnetic contribution and assuming that the ferrigmanetic contribution is exclusively maghemite with a saturation magnetization  $(M<sub>s</sub>)$  of 414  $kA/m$ ,  $32$  it is estimated that NFRPs contain approximately 0.16% maghemite, while AFRPs contained 0.09% maghemite, on a mass basis. The lack of hysteresis in the M(H) cycle indicates the ferrimagnetic component of the particle structure is in the superparamagnetic regime. Superparamagnetism at room temperature is observed for spherical maghemite particles with diameters below 10 nm.32



**Figure 2.** (A): Magnetization (emu/g) of the non-activated and activated Fe(VI) resultant particles as a function of applied magnetic field. Data collected and room temperature and normalized by mass of solids. (B): magnetization in kA/m after paramagnetic subtraction, assuming the ferromagnetic material is maghemite.

**3.2. Size and Morphology.** Sulfite activation changed the size distribution and morphology of Fe(VI) resultant particles (Figure 3). Both particle types had bimodal distributions with size features in the nanometer and micrometer size range. The intensity-weighted mean particle size  $(d_{ave})$  for nonactivated and activated Fe(VI) resultant particles were 1.72 and 2.49 µm, respectively. AFRPs were more polydisperse than NFRPs with log normal standard deviations ( $\sigma_{\text{log}}$ ) of 1.98 and 2.14 nm, respectively. AFRPs were least prominent in the  $1 - 2 \mu m$  size range, compared to nonactivated particles which were most common in that range. NFRPs resulted in  $\sim$ 90% of cumulative intensity response occurring below  $2 \mu m$ , while AFRPs resulted in ~50%. Below 2 µm diameter, the first distribution  $d_{ave}$  changes to 1091 nm and 511 nm for NFRPs and AFRPs, respectively (see Table S1). GF/F DLS results were similar, showing a difference in the submicron size range (Figure 3, panels A and C), after larger particles have been removed. On a mass basis,  $> 98\%$  of all Fe was retained on the GF/F filter, indicating that both particle types were almost all  $> 0.7 \mu m$ , and likely above the measurement range of the DLS instrument, due to limited diffusion.

Very low counts were noted below 100 nm in all GF/F DLS measurements, indicating a low concentration of nanoparticles, however, particles of that size or smaller were captured via TEM analysis (see Figure 3B and 3D), following particulate concentration from GF/F effluent on the 30 kDa membrane UF. Nanoparticle morphology was different between the NFRPs (Figure 3B) and AFRPs (Figure 3D). NFRPs TEM images

had clearly defined regions of contrast compared to AFRPs, indicating greater crystallinity, and appeared to contain aggregates of < 10 nm nanoparticles–a trend noted in prior TEM investigations, although precipitation condition varied.<sup>23,29,33</sup> SEM also showed more defined nanoparticles for the NFRP than the AFRP (see Figure SI S3 and S4). XRD results indicated, however, that NFRPs were amorphous (see Figure S2). This apparent disagreement is attributable to the size and architecture of nanoscale NFRPs and the large fraction of non-magnetic material: nanoscale maghemite may appear amorphous to XRD due to loss of symmetry near the surface of the particle, and poorly crystalline  $\gamma$ -FeOOH shells and other amorphous phases may obscure the underlying more-crystalline  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> core of NFRPS.<sup>27</sup>



**Figure 3.** Left, particle size distributions for nonactivated (A) and activated (C) ferrate resultant particles before and after a glass fiber fine (GF/F) filter. Right, transmission electron microscopy (TEM) images of nonactivated (B) and activated (D) ferrate resultant particles. Scale bar represents 50 nm in each TEM image.

**3.3. Surface Charge.** Particles resulting for both formation reactions had negative surface charges, representing a relatively stable colloidal suspension. Sulfite activation had only minor impacts on the surface charge of resultant particles (see SI-S6 and

Figure S5 for more details).

#### **4. DISCUSSION & CONCLUSION**

**4.1. Mechanistic Interpretation.** The differences in characteristics between NFRPs and AFRPs suggest differences in precipitation mechanism attributable to the presence of sulfite/sulfate. An stoichiometrically-excessive amount of sulfite will reduce Fe(VI) to Fe(III) rapidly  $(k > 10^{12} M^{-1} s^{-1})$ ,<sup>34</sup> resulting in near-instantaneous formation of Fe(III) and  $SO_4$ <sup>-2</sup>. Fe(VI) auto-decay produces Fe(III) at a rate many orders of magnitude slower at pH 7.5 ( $k \sim 20$  M<sup>-1</sup> s<sup>-1</sup>).<sup>22,35</sup> Ferric-oxide particles typically form by growth of nuclei fed by low-molecular weight iron species (e.g. dimeric iron hydroxo-species), with the ultimate form and crystallinity dedicated by the rate at which these species are supplied.36,37 The more slowly the species are supplied, as in Fe(VI) auto-decay, the better ordered phases that result.38,39 Therefore, the slower "supply" of Fe(III) resulting from nonactivated Fe(VI) would set conditions for more crystalline structures, while activation likely leads to more amorphous structures (Figure 3). This mechanistic difference could apply to any mode of Fe(VI) activation that drastically accelerates the rate of decay. The presence of sulfate resulting from sulfite-based Fe(VI) activation may also specifically impact particle precipitation mechanisms. Sulfate forms complexes with Fe(III), and may impact particle formation in several ways, including increasing the rate of precipitation compared to solutions without divalent anions.37,40

**4.2. Water Treatment Implications.** Magnetism of NFRPs has been highlighted as a potential advantage of Fe(VI) in a water treatment context, as it would allow for magnetically-based particulate separation.23,41 A force balance on maghemite particles indicates magnetic force is approximately 5 times larger than gravity for particles 0.5 cm

away from a 1 T magnet (see SI-S7). This dominance of magnetism is important, as exclusively gravimetric approaches to NFRP separation are generally effective (Figure 1),29 and predicted Stoke's terminal settling velocities for NFRPs are < 3 mm/day when diameter is < 500 nm. Attractive interparticle magnetic forces may also lead to aggregation of particles, in addition to van der Waals forces,42 which in turn could lead to enhanced gravimetric precipitation. Activation of Fe(VI) has been highlighted as a novel approach to advanced oxidation of recalcitrant organic contaminants. Results here demonstrate that a common method for Fe(VI) activation decreases subsequent particle magnetism and crystalline-morphology, while also changing the size distribution and settling velocities. In this way, Fe(VI) activation with sulfite represents a trade-off between improved oxidation and impeded downstream physicochemical processes. In addition, the crystal structure of FRPs is critical to the adsorption of arsenic and other contaminants.27,41 Our results show that sulfate activation changes the structure of the ferrate resulting particles, which could lead to a change in contaminant adsorption. Different activation approaches exist, such as sub-stoichiometric and/or staggered sulfite addition,43 which may better balance treatment goals. Other important solutes and water quality characteristics (i.e. dissolved organic carbon, pH) may impact resultant particle characteristics. Future work on activated Fe(VI) should consider the formation and fate of resultant particles in relevant water matrices, and evaluate the trade-off between improved oxidation of contaminants and reduced settleability, magnetism, and adsorption characteristics of the downstream particles.

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